

Electrical Spin Injection and Detection in $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ Nanowire Transistors

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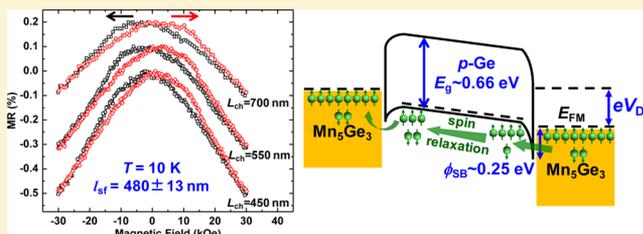
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S Supporting Information

ABSTRACT: In this Letter, we report the electrical spin injection and detection in Ge nanowire transistors with single-crystalline ferromagnetic Mn_5Ge_3 as source/drain contacts formed by thermal reactions. Degenerate indium dopants were successfully incorporated into as-grown Ge nanowires as p-type doping to alleviate the conductivity mismatch between Ge and Mn_5Ge_3 . The magnetoresistance (MR) of the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor was found to be largely affected by the applied bias. Specifically, negative and hysteretic MR curves were observed under a large current bias in the temperature range from $T = 2$ K up to $T = 50$ K, which clearly indicated the electrical spin injection from ferromagnetic Mn_5Ge_3 contacts into Ge nanowires. In addition to the bias effect, the MR amplitude was found to exponentially decay with the Ge nanowire channel length; this fact was explained by the dominated Elliot-Yafet spin-relaxation mechanism. The fitting of MR further revealed a spin diffusion length of $l_{\text{sf}} = 480 \pm 13$ nm and a spin lifetime exceeding 244 ps at $T = 10$ K in p-type Ge nanowires, and they showed a weak temperature dependence between 2 and 50 K. Ge nanowires showed a significant enhancement in the measured spin diffusion length and spin lifetime compared with those reported for bulk p-type Ge. Our study of the spin transport in the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor points to a possible realization of spin-based transistors; it may also open up new opportunities to create novel Ge nanowire-based spintronic devices. Furthermore, the simple fabrication process promises a compatible integration into standard Si technology in the future.

KEYWORDS: Spin injection and detection, Ge nanowire transistor, Mn_5Ge_3 , spin transport, spinFET



As the aggressive scaling of the physical feature size continues in the complementary metal-oxide-semiconductor (CMOS) technology, there is an urgent need to resolve a number of challenging issues. Among them, the increase in functionality, the decrease in power dissipation, and the reduction in manufacturing variability are of key challenges as highlighted in the *International Technology Roadmap of Semiconductors* (ITRS).¹ Spin-based electronics (spintronics) have emerged as a promising solution in utilizing the spin of electrons as a new degree of freedom for information processing, enabling the creation of novel electronic devices with low power consumption and nonvolatility.^{2,3} In the building of semiconductor-based spintronic devices, such as spin field-effect transistor (spinFET) and spin light-emitting diode (spinLED),^{3–5} one of the key issues is to realize an efficient spin injection from a ferromagnet (FM) into a semiconductor (SC). Electrical spin injection into various

semiconductors, including Si,^{6,7} Ge,^{8,9} GaAs,^{10,11} and graphene,^{12,13} has attracted numerous research efforts in the past two decades. While the electrical spin injection into ordinary metals can be easily demonstrated in metallic spin valve structures,^{14,15} the realization of efficient spin injection into semiconductors is much complicated by several factors: (1) the huge difference in the conductivity between ordinary ferromagnetic metals and semiconductors would almost make the spin injection efficiency negligibly small, well known as the conductivity mismatch problem;¹⁶ (2) the increase in the doping concentration in order to reduce such a conductivity difference, however, would decrease the carrier spin lifetime due to the aggravated spin relaxation originating from impurity

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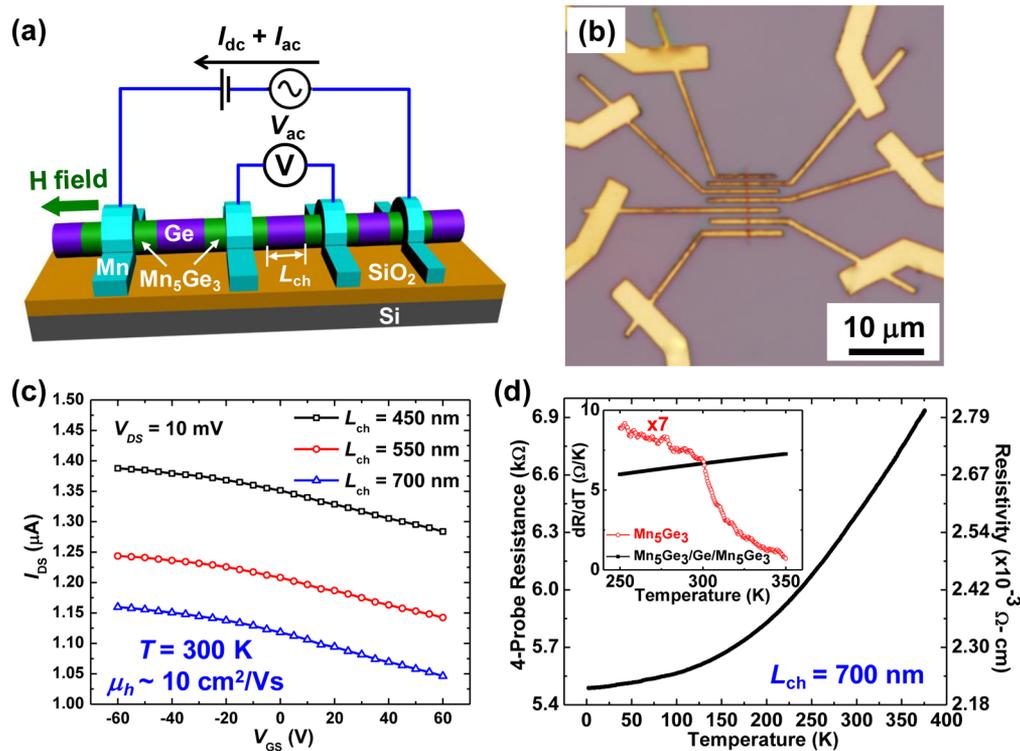


Figure 1. Characterization of single-crystalline $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors. (a) Schematic illustration of Ge nanowire transistors with thermally formed Mn_5Ge_3 as Schottky source/drain contacts. The setup for a standard 4-probe measurement is also illustrated. The magnetic field for the following measurements in this study was applied along the nanowire axial direction. (b) Optical microscope image of multiple as-fabricated $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors on a SiO_2/Si substrate. (c) $I_{\text{DS}}-V_{\text{GS}}$ curves of three Ge nanowire transistors with different channel lengths ($L_{\text{ch}} = 450, 550,$ and 700 nm) with $V_{\text{DS}} = 10$ mV, showing a p-type transistor behavior with a field-effect hole mobility of about $10 \text{ cm}^2/(\text{V s})$. (d) The 4-probe resistance-temperature (R-T) measurement on the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor with a channel length of $L_{\text{ch}} = 700$ nm, showing a decreasing resistance with the decreasing temperature. This confirms the degenerate indium doping in the Ge nanowire. The inset shows the first-order derivative of the R-T curve (black curve), showing a smooth transition near $T = 300$ K. For comparison, the data of a Mn_5Ge_3 nanowire (red curve, multiplied by a factor of 7) adopted from ref 24 is also plotted here.

scatterings;^{3,17} (3) while the insertion of a tunneling or Schottky barrier helps alleviate the conductivity mismatch,^{18,19} the preparation of a high-quality tunneling oxide without pinholes or a defects-free Schottky contact without Fermi-level pinning is not easy.^{20,21} Also, the localized states at the FM/SC interface and the surface roughness could significantly complicate and jeopardize the spin injection process.²² Therefore, the preparation of high-quality FM/SC structures is a key step toward realizing efficient spin injection into semiconductors. In an earlier work, we were able to fabricate single-crystalline $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors using a simple rapid thermal annealing (RTA) process²³ in which the formed Mn_5Ge_3 Schottky contacts maintained atomically clean interfaces with Ge nanowires and the Mn_5Ge_3 nanowire exhibited ferromagnetic orderings up to room temperature.²⁴ It should be pointed out that the Curie temperature of Mn_5Ge_3 can be further increased up to 445 K with appropriate carbon doping in order to build practical spintronic devices that can operate at room temperature.²⁵ Moreover, such one-dimensional high-quality germanide/Ge contacts formed by RTA were found to effectively alleviate the Fermi level pinning,²⁶ for which conventional metal/Ge contacts were suffered.²⁷ This should allow us to probe the intrinsic spin property in the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor. Indeed, with a high spin polarization,²⁸ Mn_5Ge_3 has been theoretically predicted to be a high-efficiency spin injection source into Ge.²⁹ In this work, we demonstrate the

spin injection into Ge nanowires through Mn_5Ge_3 source/drain contacts in $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors. We also study the bias effect and the temperature dependence of the spin transport, revealing a weak temperature dependence for the spin diffusion length of about 480 nm and the spin lifetime exceeding 244 ps in degenerately p-doped Ge nanowires. These numbers in Ge nanowires are much larger than those reported in bulk Ge with a similar doping level,³⁰ which can be attributed to the significant suppression of spin relaxation because of quantum confinements in nanostructures.

Results and Discussion. Single-crystalline $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors were fabricated on a Ge nanowire with Mn metal contacts using a RTA process, as described in a prior work for the formation of ferromagnetic source/drain contacts in Ge nanowire transistors.²³ Figure 1a schematically illustrates the device structure, in which the ferromagnetic Mn_5Ge_3 nanowire contacts were formed upon 450 °C RTA. The formed $\text{Mn}_5\text{Ge}_3/\text{Ge}$ heterostructure with an atomically clean interface provides a defects-free FM/SC structure for electrical spin injection. The Schottky barrier for the Mn_5Ge_3 contact to a p-type Ge nanowire was determined to be about 0.25 eV from temperature-dependent $I-V$ measurements.²³ To overcome the fundamental obstacle of conductivity mismatch,¹⁶ heavily doped Ge nanowires were required to achieve appreciable spin injection from the ferromagnetic Mn_5Ge_3 contact into the Ge nanowire. In the previous work, as-grown Ge nanowires were not intentionally doped during the vapor-

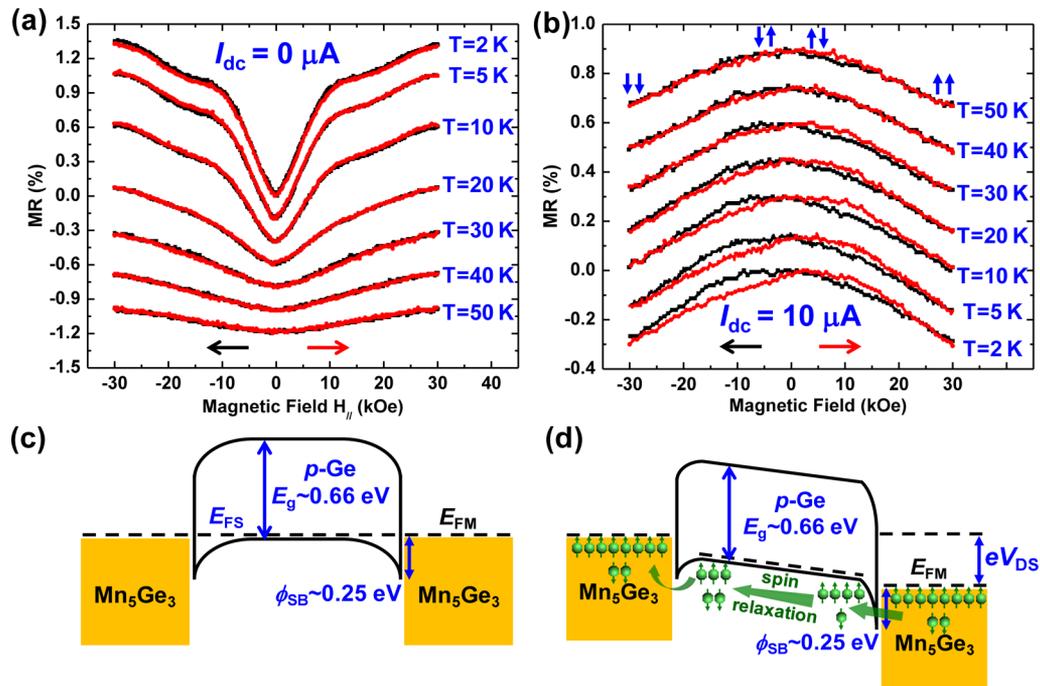


Figure 2. MR loops of a $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor with a channel length of $L_{ch} = 700$ nm at different temperatures from 2 to 50 K under a dc current bias of (a) $I_{dc} = 0 \mu A$ and (b) $I_{dc} = 10 \mu A$, respectively. The black and red arrows indicate the backward and forward sweeping directions of the axial magnetic field between -30 and 30 kOe, respectively. For clarity, MR curves in (a) and (b) are intentionally offset by multiples of 0.2 and 0.15%, respectively. The $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor showed positive MR curves with no apparent hysteresis under zero dc bias ($I_{dc} = 0 \mu A$) while negative and hysteretic MR curves under a large dc bias ($I_{dc} = 10 \mu A$). The blue arrows in (b) indicate the relative magnetization directions of the Mn_5Ge_3 spin injector and the spin detector. Energy band diagrams of the $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor under zero and large dc bias are shown in (c,d), respectively. The transport process of spin-polarized carriers (holes) in the $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor is also illustrated in (d).

liquid–solid growth process.³¹ Here in order to introduce a high doping density, as-grown Ge nanowires were exposed in an indium ambient at about $600^\circ C$ inside an ultrahigh-vacuum molecular beam epitaxy (MBE) chamber for 2 hours to drive indium atoms diffusing into Ge nanowires as p-type dopants, prior to being transferred onto SiO_2/Si substrates for device fabrication. The Ge nanowire morphology was inspected with transmission electron microscope (TEM) to ensure that the Ge nanowire maintained the cubic crystal structure after incorporating indium doping (see Figure S1 in Supporting Information). Furthermore, the energy-dispersive X-ray spectrum (EDS, with a resolution of about 0.1% atomic concentration) of indium-doped Ge nanowires revealed a very small peak of indium, affirming the successful incorporation of indium dopants into the Ge lattice (see Figure S2 in Supporting Information).

Figure 1b shows the optical microscope image of multiple as-fabricated $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistors after transferring indium-doped Ge nanowires onto a SiO_2/Si substrate. No scanning electron microscope (SEM) image was taken to avoid undesirable damages on the device by the high-energy electron beam; however it was expected to be similar to the prior work.²³ The Si substrate was degenerately doped and used as a back-gate electrode while the 300 nm thick SiO_2 layer grown on top by thermal oxidization served as the gate dielectric. Figure 1c shows the $I_{DS}-V_{GS}$ curves of three $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistors with different channel lengths ($L_{ch} = 450, 550,$ and 700 nm) under $V_{DS} = 10$ mV measured in the room-temperature ambient (see Figure S3 in Supporting Information for the detailed device

dimensions). They all showed a similar p-type transistor behavior with a field-effect hole mobility of about $\mu_h = 10$ cm²/(V s), which was calculated from the transconductance as discussed in prior works.^{32–34} It was noted that the Ge nanowire transistors showed about 1 order of magnitude improvement in the carrier mobility after RTA at $450^\circ C$ (see Figure S4 in Supporting Information), and such an improvement was attributed to the formation of high-quality Mn_5Ge_3 source/drain contacts after annealing.³³ Compared with the reported performance of $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistors built on as-grown Ge nanowires,²³ the devices here showed a much lower hole mobility and a smaller current on/off ratio while with a higher nanowire conductance. This result implied a high concentration of indium doping in the Ge nanowire due to the fact that it is difficult to deplete a heavily doped semiconductor and the high-concentration dopants drastically degrade the carrier mobility through impurity scatterings.³⁵ To quantitatively evaluate the Ge nanowire resistivity and the indium doping concentration, temperature-dependent resistance measurements were performed on a $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor with a channel length of $L_{ch} = 700$ nm and with a diameter of about 60 nm. A standard 4-probe measurement setup with a lock-in technique was used to exclude the contact resistance and improve the signal-to-noise ratio in the measurement. The measurement setup is illustrated in Figure 1a. As shown in Figure 1d, the $Mn_5Ge_3/Ge/Mn_5Ge_3$ nanowire transistor exhibited a monotonically decreasing resistance with decreasing temperature down to $T = 2$ K. This behavior reaffirmed the degenerate doping in the Ge nanowire in which case the dominated

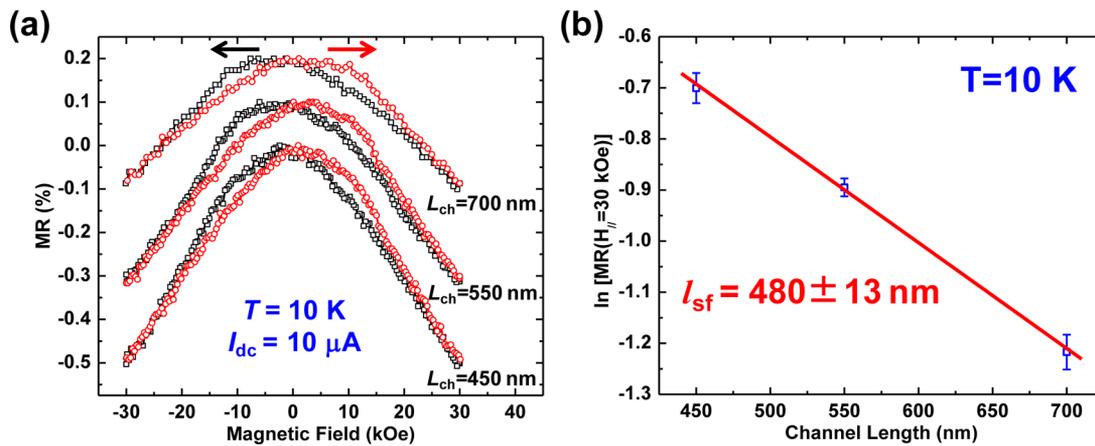


Figure 3. (a) MR curves of three $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor with three different channel lengths ($L_{\text{ch}} = 450, 550,$ and 700 nm) at $T = 10$ K under a dc current bias of $I_{\text{dc}} = 10 \mu\text{A}$. The black and red arrows indicate the backward and forward sweeping directions of the axial magnetic field between -30 and 30 kOe, respectively. All the MR curves are intentionally offset by multiples of 0.1% for clarity. (b) Semilog plot of the MR magnitude at $H_{\parallel} = 30$ kOe versus the channel length for the three $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors. The linear fitting (red curve) yields a spin diffusion length of $l_{\text{sf}} = 480 \pm 13$ nm in the p-type Ge nanowire at $T = 10$ K.

impurity scattering was effectively screened by the high-density free carriers.³⁶ The first-order derivative of the 4-probe resistance with respect to the temperature (dR/dT) was also plotted in the inset of Figure 1d, showing a smooth change from 250 to 350 K for the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor. Compared with the clear cusp in the dR/dT curve near $T_{\text{C}} = 300$ K for the Mn_5Ge_3 nanowire in ref 24 (also replotted in the inset of Figure 1d for comparison),²⁴ this result suggested that the part of the resistance from the Mn_5Ge_3 source/drain was small and most of the measured resistance came from the Ge nanowire channel. Quantitatively, using the previously measured resistivity value for the Mn_5Ge_3 nanowire,²⁴ we calculated that the Mn_5Ge_3 contacts only contributed to less than 3% of the total $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ resistance. Therefore, for simplicity, we assumed the measured total resistance equaled to the Ge nanowire channel resistance. Then the resistivity of the Ge channel was calculated to be $\rho_{\text{Ge}} = 2.58 \times 10^{-3} \Omega\text{-cm}$ at $T = 300$ K and decreased slightly to $\rho_{\text{Ge}} = 2.22 \times 10^{-3} \Omega\text{-cm}$ at $T = 2$ K. On the other hand, the previously reported resistivity values for the Mn_5Ge_3 nanowire were $\rho_{\text{FM}} = 2.4 \times 10^{-4} \Omega\text{-cm}$ at $T = 300$ K and decreased to $\rho_{\text{FM}} = 4.65 \times 10^{-5} \Omega\text{-cm}$ at $T = 2$ K.²⁴ The conductivity ratio (inversely the resistivity ratio), $\sigma_{\text{FM}}/\sigma_{\text{Ge}} = \rho_{\text{Ge}}/\rho_{\text{FM}}$ of the $\text{Mn}_5\text{Ge}_3/\text{Ge}$ heterostructure was then evaluated to be between 10 and 50 in the temperature range of 2 – 300 K. This value is more than 20 times smaller than that of ordinary FM/Ge (e.g., Fe/Ge, Co/Ge) spin injection structures, given that the resistivity of commonly used ferromagnetic metals is typically on the order of 10^{-6} to $10^{-5} \Omega\text{-cm}$, or at least 20 times lower than that of Mn_5Ge_3 .³⁷ Therefore, compared with ordinary FM/Ge structures, the conductivity mismatch in the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor should be significantly reduced to facilitate the spin injection into Ge. It is also worth mentioning that the spin polarization in Mn_5Ge_3 was experimentally reported to be about $P_{\text{FM}} = 42\%$ at $T = 1.2$ K,²⁸ which is comparable with that of conventional ferromagnetic metals (for instance, $P_{\text{Fe}} = 45\%$, $P_{\text{Co}} = 42\%$, and $P_{\text{Ni}} = 33\%$).³⁸ The relatively low conductivity and high spin polarization of Mn_5Ge_3 , along with the atomically clean interfaces with Ge nanowires, suggests that Mn_5Ge_3 is a promising ferromagnetic material for spin injection into Ge nanowires.

In the characterization of spin injection and transport, Hanle precession measurement is widely used to extract the spin lifetime and spin diffusion length.¹⁵ Usually nonlocal measurements with a four-terminal geometry are adopted to detect the intrinsic spin injection signal in order to avoid any spurious signals.³⁹ Unfortunately, this technique could not be used for our $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor (see Figure S5 in Supporting Information for detailed explanation). Alternatively, we may use the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor as a vertical spin valve with a current perpendicular-to-plane configuration.^{40,41} In this case, spin-polarized carriers are injected into the Ge nanowire from one ferromagnetic Mn_5Ge_3 contact (namely the spin injector) and then are scattered as they travel along the Ge nanowire channel before reaching the other ferromagnetic Mn_5Ge_3 contact (namely the spin detector). This process will be discussed in details later (illustrated in Figure 2d). For nonpolar semiconductors like Ge, the dominant spin relaxation mechanism is the Elliot-Yafet spin flip mechanism that occurs when scattered with phonons and impurities.³ In experiment, a constant dc bias current superimposed with a small ac current of $I_{\text{ac}} = 1 \mu\text{A}$ was flowed through the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor, and the ac voltage signal was sensed with standard lock-in technique while sweeping the axial magnetic field. The easy-axis of the Mn_5Ge_3 nanowire was found to be along the nanowire axis, and the two ferromagnetic Mn_5Ge_3 contacts were intentionally designed to have different lengths and hence possibly different coercive fields. As the axial magnetic field was swept back and forth between -30 and 30 kOe, the relative magnetization directions of the spin injector and the spin detector were changed between parallel and antiparallel configurations. Figure 2a,b shows the MR of a $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor with a channel length of $L_{\text{ch}} = 700$ nm under two different dc current biases in the temperature range from 2 to 50 K. The MR here is defined as $\text{MR} = ((R(H) - R_{\text{min}})/R_{\text{min}}) \times 100\%$ (positive MR) for Figure 2a, and $\text{MR} = ((R(H) - R_{\text{max}})/R_{\text{max}}) \times 100\%$ (negative MR) for Figure 2b. It is interesting to observe that the MR curves under a zero and a high dc bias current showed distinct characteristics: positive MR with no apparent hysteresis with $I_{\text{dc}} = 0 \mu\text{A}$ while there is a negative and hysteretic MR under $I_{\text{dc}} = 10 \mu\text{A}$. It should be pointed out that while the positive MR under $I_{\text{dc}} = 0 \mu\text{A}$ is

likely attributed to the longitudinal MR of Ge,⁴² the bias-dependent MR characteristics could not originate from the Ge nanowire, the Mn₅Ge₃ contact, and associated anisotropy magnetoresistance (see Figure S6 in Supporting Information for detailed explanation). The bias effect on the MR behavior can be simply explained from the energy band diagram schematically shown in Figures 2c,d: since there is a Schottky barrier height of about 0.25 eV for the ferromagnetic Mn₅Ge₃ contact to the p-type Ge nanowire,²³ a large enough dc bias voltage (current) is required to reduce the Schottky barrier width in the reverse biased spin injector terminal to allow for sufficient spin-polarized carriers being injected into the Ge nanowire channel and moving toward the spin detector. The transport process of spin-polarized carriers in the Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistor is also schematically illustrated in Figure 2d. As shown in Figure 2b, with a dc current bias of $I_{dc} = 10 \mu\text{A}$ we were able to observe the negative and hysteretic MR from $T = 2 \text{ K}$ up to $T = 50 \text{ K}$, unambiguously demonstrating the spin injection and detection in the Ge nanowire transistor. More MR curves are shown in the Supporting Information (see Figure S7 in Supporting Information). It is worth noting that similar bias effect on the MR behavior was also observed in the MnSi/Si/MnSi nanowire heterostructure in which spin-polarized carriers were injected from the Schottky MnSi contact into the p-type Si nanowire.⁴³

To quantitatively determine the spin diffusion length, temperature-dependent MR measurements were performed on several Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistors with different channel lengths. Figure 3a shows the MR curves of three Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistors with three different channel lengths ($L_{ch} = 450, 550, \text{ and } 700 \text{ nm}$) at $T = 10 \text{ K}$ under the same dc current bias of $I_{dc} = 10 \mu\text{A}$. They showed similar negative and hysteretic characteristics with a systematic decrease in the MR magnitude with the increasing Ge nanowire channel length. Recalling the Julliere's model for a FM/insulator/FM structure,⁴⁴ the tunneling magnetoresistance (TMR) is given by

$$\text{TMR} = \frac{2P_1P_2}{1 + P_1P_2} \quad (1)$$

Here P_1 and P_2 are the spin polarizations of the two FM electrodes defined as $P = (N_{\uparrow} - N_{\downarrow}) / (N_{\uparrow} + N_{\downarrow})$ in which N_{\uparrow} (N_{\downarrow}) is the density of states at the Fermi level for the majority (minority) spin direction. In our Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire structure with a fairly long Ge channel, the Julliere's model of TMR can be modified as follows to include the spin relaxation in the Ge nanowire⁴¹

$$\text{TMR} = \frac{2P_1P_2e^{-L_{ch}/l_{sf}}}{1 + P_1P_2e^{-L_{ch}/l_{sf}}} \quad (2)$$

where L_{ch} is the Ge nanowire channel length, and l_{sf} is the spin diffusion length (see Figure S8 in Supporting Information for more discussions on the model). In the case where $P_1P_2e^{-L_{ch}/l_{sf}} \ll 1$, or equivalently the TMR magnitude is small (such as in this work), eq 2 can be further simplified as an exponential function

$$\text{TMR} \approx 2P_1P_2e^{-L_{ch}/l_{sf}} \quad (3)$$

In the MR curves of Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistors shown in Figure 3a, it is difficult to accurately determine the TMR magnitude, because typical abrupt resistance steps were not observed as in previous organic spin valves that comprised

two different ferromagnetic contacts.^{40,41} This could be attributed to a possibly small difference in the coercive field for the Mn₅Ge₃ spin injector and detector (with the same diameter but slightly different lengths). In addition, the presence of multiple domains in the ferromagnetic Mn₅Ge₃ nanowire would also prevent the abrupt switching of its magnetization.²⁴ Further investigation may be required to understand the detailed mechanism. Here, in order to extract the spin diffusion length in the Ge nanowire, we obtained the MR amplitude of three Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistors with different L_{ch} at the same temperature and magnetic field. Here we picked the largest magnetic field in our measurements to ensure the magnetization of all the Mn₅Ge₃ nanowires in parallel. The rationale is to compare the MR ratio of three Mn₅Ge₃/Ge/Mn₅Ge₃ nanowire transistors with different Ge nanowire channel lengths while keeping all other parameters the same (same temperature and bias condition, etc.). Therefore, the change in the MR ratio should closely related to the spin relaxation in the Ge nanowire. A semilog plot of the MR magnitude at $H_{\parallel} = 30 \text{ kOe}$ versus the channel length is then plotted at $T = 10 \text{ K}$ in Figure 3b. Using eq 3, one yields a spin diffusion length of $l_{sf} = 480 \pm 13 \text{ nm}$ in the p-type Ge nanowire at $T = 10 \text{ K}$. To make a meaningful comparison with previous results in Ge, we first estimated the doping concentration in our Ge nanowires. Using the resistivity-doping concentration relation in bulk Ge as an approximation,⁴⁵ the resistivity value of $\rho_{\text{Ge}} = 2.58 \times 10^{-3} \Omega\text{-cm}$ at $T = 300 \text{ K}$ corresponds to a p-type doping concentration of about $N_A = 8 \times 10^{18} \text{ cm}^{-3}$. Measurements from more than 10 devices yielded a p-doping concentration in the range between 6×10^{18} and $9 \times 10^{18} \text{ cm}^{-3}$. From the literature,³⁰ p-type bulk Ge with a similar doping concentration of $N_A = 8.2 \times 10^{18} \text{ cm}^{-3}$ was reported to have a spin diffusion length of about $l_{sf} = 80 \text{ nm}$ (calculated at $T = 300 \text{ K}$ but it is also valid for temperature down to $T = 5 \text{ K}$ as both the spin lifetime τ_{sf} and the diffusion constant D_h have a weak temperature dependence in heavily doped semiconductors^{17,30,46}). The significant enhancement in the spin diffusion length in the one-dimensional Ge nanowire channel could be attributed to the effective suppression of electron-phonon scattering and thus spin relaxation by quantum confinements in nanostructures with a reduced density of states.⁴⁷ From eq 3, we can also extract the spin polarization of Mn₅Ge₃ to be about 8% at $T = 10 \text{ K}$, which is much smaller than the reported $(42 \pm 5)\%$ at $T = 1.2 \text{ K}$ from point contact Andreev reflection measurements.²⁸ This large deviation is mainly because that the simplified TMR model here does not include the spin injection efficiency from Mn₅Ge₃ into Ge (see Figure S8 in Supporting Information for more discussions).

To estimate the spin lifetime, we first calculated the diffusion constant, which can be obtained in degenerately doped semiconductors using the full Fermi-Dirac expression³⁵

$$D_h = \frac{2\mu_h k_B T}{q} \frac{F_{1/2}(\eta_F)}{F_{-1/2}(\eta_F)} \quad (4)$$

in which μ_h is the hole mobility, k_B is the Boltzmann constant, q is the electron charge, $\eta_F = (E_V - E_F) / k_B T$ while $F_{1/2}(\eta_F)$ and $F_{-1/2}(\eta_F)$ are the Fermi-Dirac integrals. The Fermi-level position in degenerate semiconductors can be determined using the Joyce-Dixon approximation⁴⁸

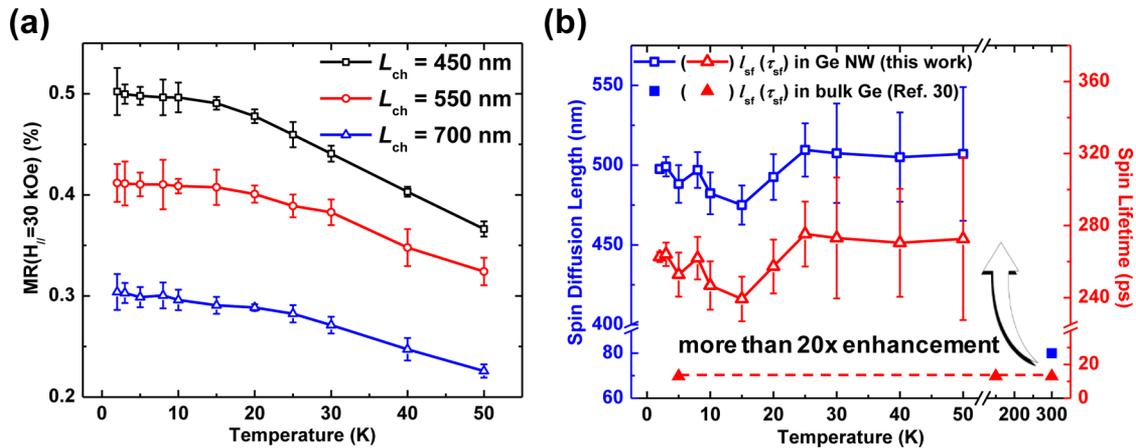


Figure 4. (a) Temperature-dependent MR magnitudes at $H_{||} = 30$ kOe for three $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors with different channel lengths ($L_{\text{ch}} = 450, 550,$ and 700 nm) under a dc current bias of $I_{\text{dc}} = 10 \mu\text{A}$, showing a similar temperature dependence. (b) Fitted temperature-dependent spin diffusion length and calculated spin lifetime, showing a weak dependence on the temperature from 2 to 50 K. The apparent dip at 10 K is likely due to the fitting error. For comparison, the data from bulk Ge with a similar p-type doping concentration in ref 30 are also included and represented by solid symbols. Ge nanowires showed more than 20 times enhancement in the spin diffusion length and spin lifetime compared with bulk Ge.

$$E_V - E_F \approx k_B T \left(\ln \frac{N_A}{N_V} + 2^{-1.5} \times \frac{N_A}{N_V} \right) \quad (5)$$

Using the measured field-effect hole mobility of $\mu_h = 10 \text{ cm}^2/(\text{V s})$ at $T = 300 \text{ K}$ we can calculate the diffusion constant to be $D_h = 0.377 \text{ cm}^2/\text{s}$ given $N_A = 8 \times 10^{18} \text{ cm}^{-3}$ and $N_V = 6 \times 10^{18} \text{ cm}^{-3}$.³⁵ As the diffusion constant is weakly dependent on the temperature,^{17,30} we assume the same value for low temperatures so we can further estimate the spin lifetime at $T = 10 \text{ K}$ using

$$\tau_{\text{sf}} = \frac{l_{\text{sf}}^2}{D_h} = 6.11 \text{ ns} \quad (6)$$

It is recognized that the mobility value extracted from the $I_{\text{DS}}-V_{\text{GS}}$ curves may underestimate the conductivity mobility considering the back-gated device structure and the round shape of Ge nanowires.²³ Therefore, the above-calculated $\tau_{\text{sf}} = 6.11 \text{ ns}$ is the upper limit for the spin lifetime. If we adopt the reported hole mobility of $\mu_h = 250 \text{ cm}^2/(\text{V s})$ from a p-type Ge thin film with a similar doping concentration,³⁰ eq 6 would yield the lower limit for the spin lifetime of about $\tau_{\text{sf}} = 244 \text{ ps}$ with $D_h = 9.43 \text{ cm}^2/\text{s}$. Still this value is 1 order of magnitude larger than that observed in bulk Ge,³⁰ again implying the advantage of low-dimensional nanostructures in the building of spintronic devices with long spin lifetime and spin diffusion length.⁴⁷

Furthermore, to investigate the temperature dependence of the spin diffusion length, we obtained the temperature-dependent MR magnitudes at $H_{||} = 30$ kOe for three $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors with different channel lengths ($L_{\text{ch}} = 450, 550,$ and 700 nm) as shown in Figure 4a. Their MR magnitudes showed a similar temperature dependence: the MR magnitude linearly increases with reducing temperature from $T = 50 \text{ K}$ and then gradually saturates as the temperature went below $T = 10 \text{ K}$. Using eq 3, we obtained the temperature-dependent spin diffusion length as given in Figure 4b, showing a weak temperature dependence of the spin diffusion length in the range of $T = 2 \text{ K}$ to $T = 50 \text{ K}$. The calculated spin lifetime at each temperature using eq 6 is also plotted, assuming a constant diffusion constant of $D_h =$

$9.43 \text{ cm}^2/\text{s}$. Such a weak temperature dependence of the spin diffusion length and lifetime was also observed in heavily doped Si thin films.^{17,46} This fact may be explained by the effectively screened ionized impurity scatterings in the heavily doped Ge nanowire,³⁶ resulting in a weak dependence on the temperature for the momentum relaxation and hence the spin relaxation (as manifested by the spin diffusion length and the spin lifetime). For comparison, the data from Ge thin films with a similar p-type doping concentration in ref 30 are also included in Figure 4b, manifesting the significant enhancement of the spin diffusion length and lifetime in Ge nanostructures.⁴⁷

Conclusions. To sum up, we have successfully demonstrated electrical spin injection and detection in single-crystalline $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors fabricated on degenerately indium-doped Ge nanowires. Under zero current bias, the $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistor showed positive and symmetric MR characteristics with no apparent hysteresis; on the other hand, negative and hysteretic MR characteristics were observed under a large voltage (current) bias from $T = 2 \text{ K}$ up to $T = 50 \text{ K}$. The hysteretic MR signature clearly indicated spin injection from the ferromagnetic Mn_5Ge_3 contact into the Ge nanowire, and the large bias helped reduce depletion region width of the $\text{Mn}_5\text{Ge}_3/\text{Ge}$ junction to increase the spin injection efficiency. Furthermore, based on the modified Julliere's model, the MR of three $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors gave a spin diffusion length of $l_{\text{sf}} = 480 \pm 13 \text{ nm}$ and a long spin lifetime exceeding 244 ps in p-type Ge nanowires at $T = 10 \text{ K}$. The estimated spin diffusion length is significantly larger than the channel length of state-of-the-art MOS transistors¹ and has a weak temperature dependence. The long spin diffusion length and its weak temperature dependence were explained by the dominant Elliot-Yafet spin-relaxation mechanism as the result of the impurity scatterings in Ge, which were effectively screened in degenerately doped semiconductors. These observed spin diffusion length and spin lifetime in one-dimensional Ge nanowires were much larger than those reported from bulk Ge,³⁰ which implied that the spin relaxation can be effectively suppressed in the one-dimensional channel because of the quantum confinement effect.^{9,47}

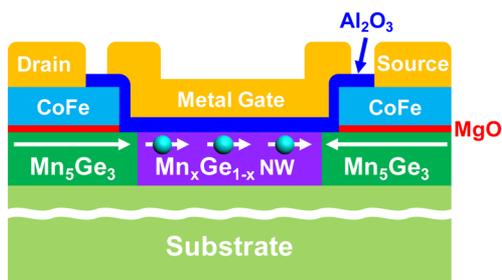


Figure 5. Schematic illustration of a nonvolatile transpinor device, which is built on a $\text{Mn}_x\text{Ge}_{1-x}$ DMS nanowire with ferromagnetic Mn_5Ge_3 contacts. Another choice for the source/drain contacts could be $\text{Co}(\text{Fe})/\text{MgO}$ tunnel junctions. The $\text{Co}(\text{Fe})/\text{MgO}$ magnetic tunnel junctions are used to read out/manipulate the magnetization of the Mn_5Ge_3 nanomagnets. Other strategies could involve spin valve structures or the inverse spin Hall effect. The ferromagnetism of the $\text{Mn}_x\text{Ge}_{1-x}$ DMS nanowire channel can be controlled by the gate electric field. The magnetic moments are transferred from the source to the channel and then to the drain.

With a relatively long spin diffusion length and the convenient fabrication process for $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors, it is possible to integrate Ge nanowire-based spintronic devices into standard CMOS technology. With spin-based transistors, this may help create novel functional devices with low power dissipation and fast switching. One promising prototype, as shown in Figure 5, is to integrate a diluted magnetic semiconductor (DMS) nanowire, whose ferromagnetism can be effectively controlled by a gate electrode, with the high-quality ferromagnetic contacts for spin injection demonstrated in this work. Another variant of the ferromagnetic source/drain contacts could be $\text{Co}(\text{Fe})/\text{MgO}$ tunnel junctions for spin injection.⁹ Our previous work has successfully demonstrated the electric field control of ferromagnetism in Mn-doped Ge DMS quantum dots up to 300 K,^{49,50} and similar effect is expected in $\text{Mn}_x\text{Ge}_{1-x}$ DMS nanowires. In this spin-based transistor (transpinor), the information is stored as the magnetization of the ferromagnetic Mn_5Ge_3 contacts (nanomagnets), and it can be read out and manipulated by a magnetic tunnel junction that converts the spin signal into a current/voltage signal. Other conversion strategies may involve spin valve structures or the inverse spin Hall effect.⁵¹ The magnetic moments are transferred from the source to the channel and then to the drain through spin injection and exchange interaction. As the electric field exerted by the gate electrode manipulates the ferromagnetism in the DMS channel, the gate controls the communication between the source/drain terminals. This DMS nanowire-based transpinor points out a possible realization of all-spin logic devices with built-in memory for low-power operations.⁵²

■ ASSOCIATED CONTENT

📄 Supporting Information

TEM image and diffraction pattern of indium-doped Ge nanowires; EDS spectrum of indium-doped Ge nanowires; schematic of device dimensions; $I_{\text{DS}}-V_{\text{GS}}$ curves of $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors before and after RTA; discussion on nonlocal measurements; temperature-dependent MR curves of $\text{Mn}_5\text{Ge}_3/\text{Ge}/\text{Mn}_5\text{Ge}_3$ nanowire transistors; and discussion on Mn_5Ge_3 spin polarization. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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